

CHAPTER V

EFFECT OF ANNEALING TEMPERATURE ON THE STRUCTURAL AND OPTICAL PROPERTIES OF CuI NANOPARTICLES

5.1 Introduction

Copper iodide (CuI), a p-type direct bandgap semiconductor, has attracted the research community for many years because of its wide bandgap, negative spin-orbit splitting and high ionicity (Naomi Yamada et.al, 2016). Due to the possibility of tuning hole density with iodine doping, high hole mobility ($43.9 \text{ cm}^2\text{V}^{-1}\text{s}^{-1}$) and high transmittance in visible region, CuI is a promising candidate to meet the present need for transparent p-MOS towards the realization of transparent CMOS technology. Attesting to the eco-friendliness of the material, CuI filled single- and multi-walled CNTs have the potential to be used as biocompatible thermometers and also as a source of dietary iodine in table salt (S. Gao et.al 2010, A. Klapars et.al 2002, W.Y. Fan 2007 and Y.Y. Xu 2009). CuI shows three polymorphic forms namely (i) γ -CuI, in zinc blende structure, which exists below 370°C , (ii) β -CuI, in wurtzite structure, is observed between the temperatures 370 to 400°C and (iii) α -CuI, in rock salt structure, is the high temperature phase which exists above 400°C (M. Grundmann et.al 2016, JB. Boyce et.al 1981). As a p-type transparent semiconductor, CuI is an attractive candidate for transparent solid state lighting devices due to high exciton binding energy (62 meV) and also for transparent thermoelectric devices owing to its high thermoelectric conversion efficiency (Bruno Miguel et.al, 2018). To synthesize CuI nanostructures, many synthetic techniques have been reported in literature like reactive sputtering, pulsed laser deposition, laser assisted molecular beam deposition, polymer assisted reaction, iodination, wet chemical synthesis and sol-gel technique (WMKP. Wijekoon et.al 1993, VY. Bokshis et.al 2004, K. Reichld et.al 1978, PM.Sirimanne et.al 2003, M. Yang et.al 2004, Y. Xu et.al 2007, LR. Meng et.al 2010). In the present

investigation, copper iodide nanoparticles were prepared by conventional method and the prepared samples were annealed at four different temperatures 100, 200, 300 and 400°C and the change in structural and optical properties were studied using appropriate characterization techniques.

5.2. Experimental Methods

5.2.1 Materials

All the chemicals used in our experiments were of analytical grade and were used without any further purification. Copper (II) chloride dihydrate ($\text{CuCl}_2 \cdot 2\text{H}_2\text{O}$) procured from Merck, potassium iodide and Sodium thiosulfate from Himedia were used in our experiments. Deionized water was used throughout the experiment.

5.2.2 Synthesis of CuI Nanoparticles

In a typical synthesis, 100 mL of 0.5 M reducing agent sodium thiosulphate solution was added in drops to 100 mL of 0.5 M aqueous solution of copper chloride with constant stirring. To the stirred solution, aqueous solution of potassium iodide is added drop by drop and a brown colour precipitate is formed, which was filtered, washed and dried in oven at 60°C for 4 hours. The samples were annealed in a muffle furnace at four different temperatures of 100, 200, 300 and 400°C in ambient atmosphere for 1 hour.

5.2.3 Characterization Techniques

The X-ray diffractogram of CuI powders were recorded using XPERT-PRO diffractometer system with CuK_α radiation of wavelength 1.5406 Å in the 2θ range 20-80°. The FTIR spectra for the prepared CuI nanoparticles were recorded using Perkin Elmer spectrometer in the wavenumber range of 400-4000 cm^{-1} . The SEM images were acquired using JEOL JSM-6390 electron microscope at the accelerating voltage of 20kV. UV-visible diffuse reflectance spectra were measured at room temperature employing Agilent Cary 500 UV-VIS-NIR spectrophotometer in the wavelength range of 300-1100 nm.

Photoluminescence spectra of the CuI nanoparticles were recorded at room temperature using Shimadzu RF-6000 spectrofluorophotometer at the excitation wavelength of 320 nm in the wavelength range of 350 - 750 nm.

5.3. Results and Discussion

5.3.1 Structural Analysis

When X-rays was allowed to incident on a crystalline sample, they are diffracted in a pattern as dictated by the characteristic of the crystal structure. The Bragg's law imposes the condition for constructive interference to obtain XRD patterns and the powder diffraction patterns of the prepared samples are shown in the Fig 5.1.

The peaks in the XRD pattern of nanoparticles are in good agreement with the reported standards (JCPDS card no. 76-0207) and the peaks at 2θ values 25.5, 29.5, 42.2, 50, 61.2, 67.4, 69.4 and 77.1 could be indexed to the (111), (200), (220), (311), (222), (400), (331) and (420) planes of γ -CuI for the samples annealed at 100 and 200°C which is in accordance with that of the pure phase of copper iodide in zinc blende crystal structure.

The XRD diffractogram of the samples annealed at 300 and 400°C shows the reflections for monoclinic copper oxide along with those obtained for copper iodide. The copper oxide peaks begins to emerge around 300°C and dominates in the sample annealed at 400°C. This agrees well with the standard values reported in the literature (JCPDS-ICDD 05-0661, 13-0420, 45-0937, 48-1548) and it shows monoclinic structure of CuO. The additional peaks appear in the diffractogram of samples annealed above 300°C at 2θ values 32, 35, 38, 46, 48, 53, 58, 61, 66 and 68 could be indexed to (110), (002) & (-111), (200)& (111), (-112), (-202), (020), (202), (-113), (022), and (220) planes of CuO. The crystallite size of CuI was determined using the Scherrer formula as given by,

$$D = k \lambda / \beta \cos \theta$$

where $k = 0.9$ is the correction factor for particle shape, β is the full width at half maximum (FWHM), λ is the wavelength of X-rays and θ is the angle of incidence (B.D. Cullity, 1997). The average crystallite size of CuI nanoparticles are 54 nm, 45 nm, 43 nm and 42 nm for the samples annealed at 100, 200, 300 and 400°C respectively. The decrease in crystallite size with increasing annealing temperature could be attributed to the nucleation of monoclinic CuO in zinc blende CuI matrix with small crystallite size than that of CuI.

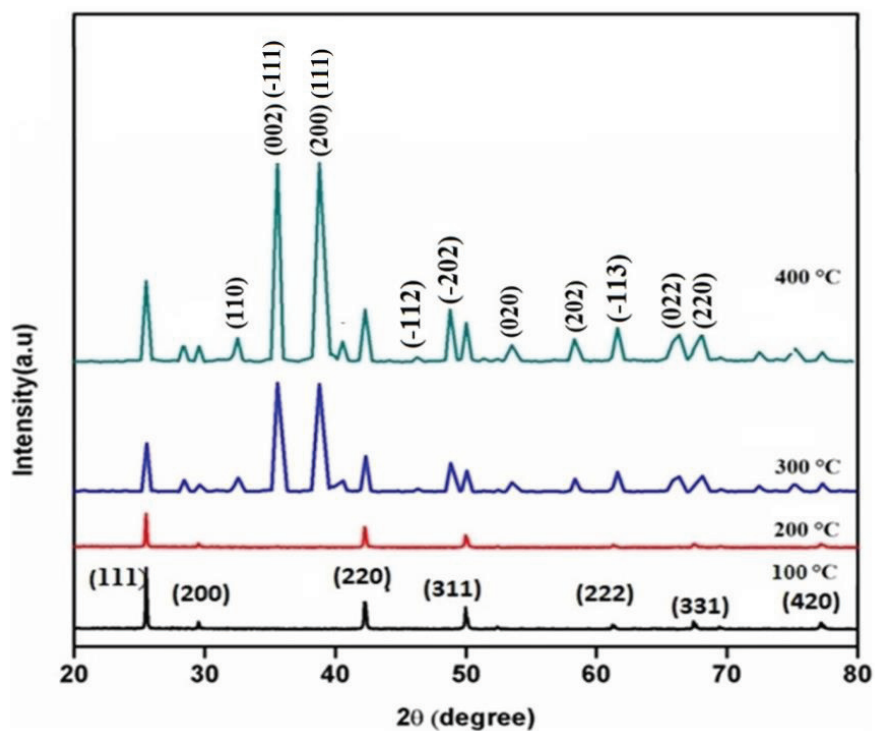


Fig 5.1 Powder XRD patterns of pure CuI and CuI-CuO mixed phase nanoparticles at different annealing temperatures

5.3.2 Functional Group Analysis

FT-IR technique is one of the most essential and extensively used spectroscopic techniques to identify the functional groups, to determine the internal structure of the molecules and to understand the nature of chemical environment in the given compound. Since the X-ray diffraction patterns of the CuI samples annealed at 100 and 200°C are found to be the same, the IR interferogram of both the samples were also similar.

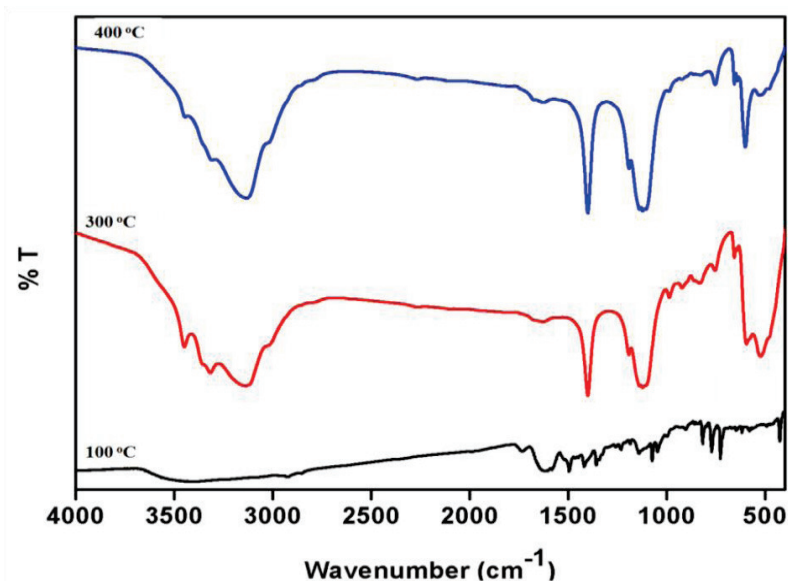
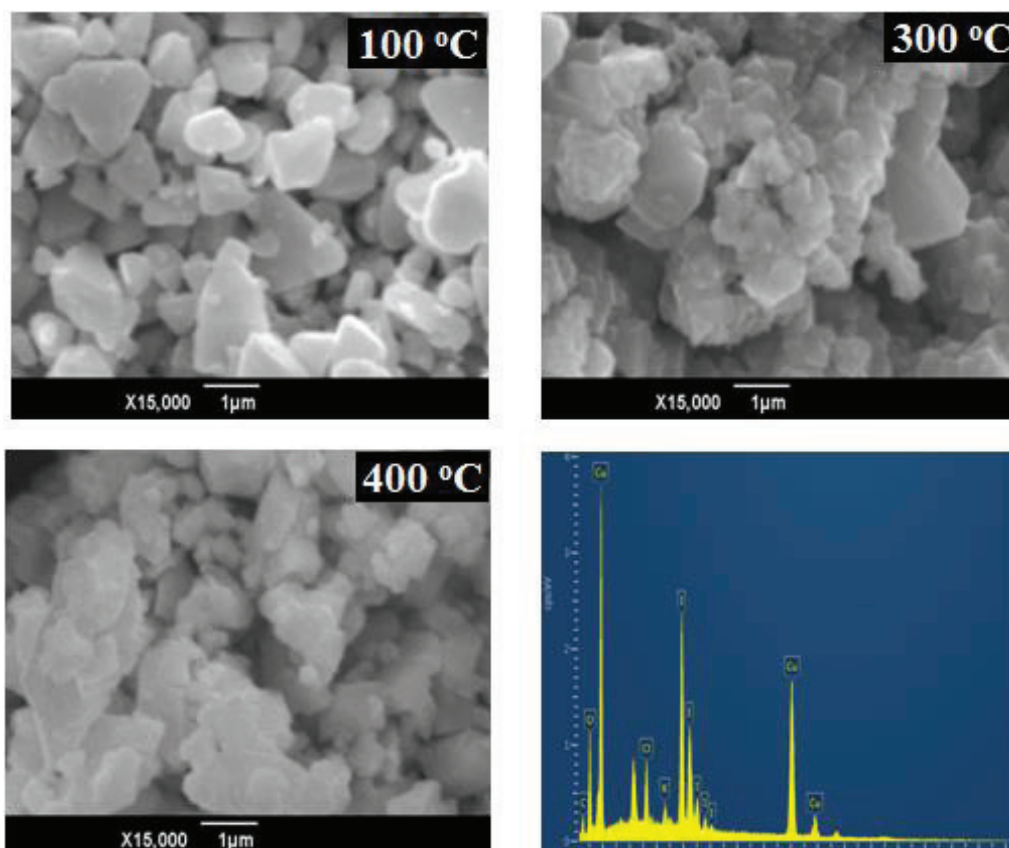


Fig 5.2 FT-IR spectra of pure CuI and CuI-CuO mixed phase nanoparticles at different temperatures

The recorded IR interferogram for the samples annealed at 100, 300 and 400°C are displayed in Fig 5.2. The bands in the range of 3450-3100 cm⁻¹ correspond to the OH stretching vibrations which may be due to the adsorbed H₂O molecules on the CuI-CuO samples. The bands were absent in pure CuI sample due to its hydrophobic nature which is further supported by the high contact angle for water (R.N. Bulakhe, 2013). The assignment of the peak observed in the region of 522 cm⁻¹ is designated to Cu-I stretching vibration (H.R. Humud et.al, 2017). The bands corresponding to copper oxide are found in the range of 500-700 cm⁻¹ (V.V. Padil and M. Cernik, 2013), which confirm the formation of CuI/CuO nanocomposites at higher annealing temperatures.

5.3.3 Morphological Analysis

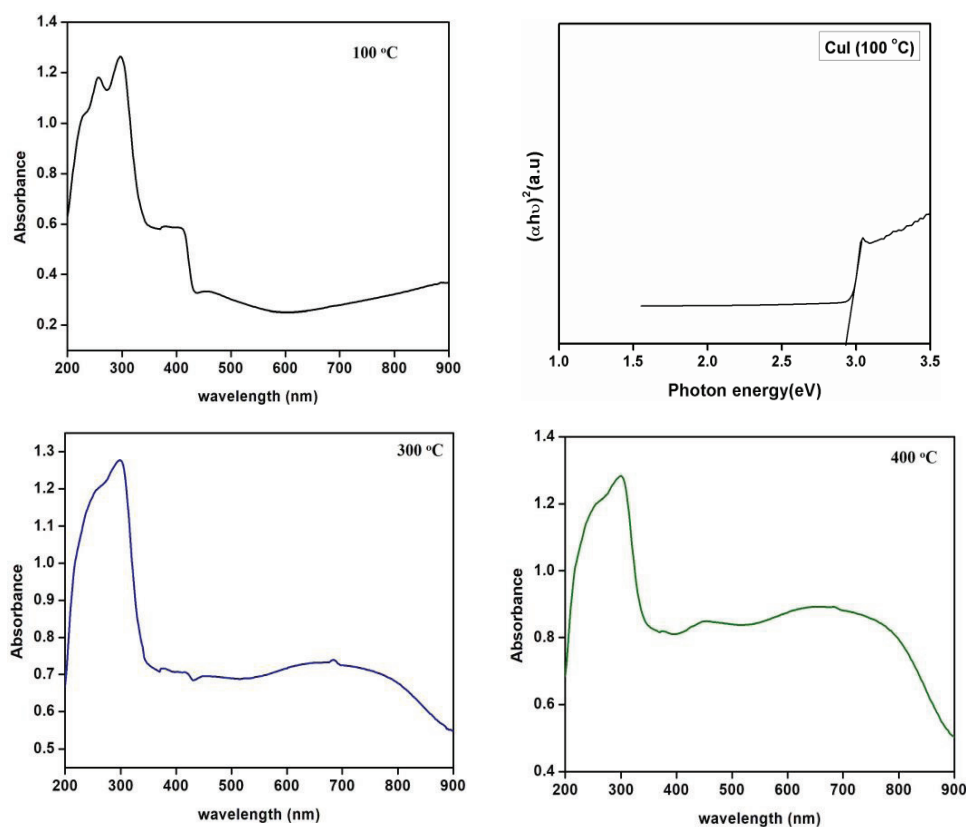
The SEM images acquired for the CuI samples annealed at different temperatures are displayed in Fig 5.3 and the significant change in morphology could be observed when the samples annealed at higher temperature. The SEM micrograph of CuI annealed at 100°C shows the triangular morphology of the sample as reported in the literature (C.Yang et.al 2016, B. Mahdi et.al 2014). The EDAX microanalysis of CuI-CuO mixed phase nanostructures (in Fig 5.3 (d)) reveals the presence of elements Cu and I as that for CuI nanostructures and in addition it also showed the presence of oxygen.



**Fig 5.3 SEM micrographs of the samples annealed at a) 100°C, b) 300°C and c) 400°C
d) EDAX spectrum of CuI-CuO mixed phase nanostructures.**

5.3.4 Absorbance Spectral Analysis

UV-visible absorption spectra of the samples were recorded in the range of 200-900 nm and are shown in the Fig 5.4(a-d). Fig5.4a shows absorption features at 320 nm and 410 nm, which could be attributed to the excitation of electrons to the conduction band from sublevels and interband excitonic absorptions respectively. But in the samples annealed at 300 and 400°C (Fig5.4c and 5.4d), there exists an additional hump in the visible region around 675 nm which is ascribed to the absorption of CuO nanoparticles (J. Koshy et.al, 2011) present in the sample.



**Fig 5.4 (a), (c), (d) UV-vis spectra of the samples annealed at 100°C, 300°C and 400°C
(b) Variation of $(\alpha h\nu)^2$ with photon energy for CuI nanoparticles annealed at 100°C**

The optical bandgap E_g is estimated using the formula,

$$(\alpha h\nu)^2 = A(h\nu - E_g)$$

where A is a constant and $h\nu$ is the photon energy. From the plot of $(\alpha h\nu)^2$ vs $h\nu$, the bandgap is determined by drawing a tangent to the linear portion of the curve and extrapolating the same to the photon energy (X -axis) at $\alpha h\nu=0$. Fig 5.4(b) shows the Tauc plot for the CuI nanoparticles annealed at 100°C, where the bandgap was found to be 2.9 eV and is in accordance with the values reported earlier for CuI nanoparticles (W. Sekkal et.al, 2002). But for samples annealed at 300 and 400°C, the absorption edge were observed at 364 nm and 354 nm respectively. The absorption features of the CuI nanoparticles are masked by the absorption spectrum of the resulting CuO formed during annealing above 400°C due to the higher absorption coefficient of CuO (C. Yang et.al 2016, A. Bhaumiket.al 2014) compared to CuI at the visible region.

5.3.5 Emission spectral analysis

The photoluminescence (PL) spectra of the samples annealed at four different temperatures are displayed in Fig 5.5. The photoluminescence spectra of CuI sample annealed at 100°C and 200°C were found to be identical. Photoluminescence of CuI sample annealed at 100°C shows a narrow blue emission peak centered at 430 nm (S.A. Lanje et.al 2010, K.O. Ighodalo, 2017) while the samples annealed at 300 and 400°C show a broad and asymmetric emission peak around 455 nm accompanied by the weak emission at 684 nm. The emission peaks at 430 nm and 455 nm originates from band edge emission of CuI nanoparticles whereas the red luminescence at 684 nm could be due to defect emission as a result of the iodine vacancies (P. Gao, 2009) formed in CuI samples with increase in annealing temperature. No emission peaks from CuO was observed in the spectrum due to apparently poor photoluminescence efficiency of CuO compared to CuI.

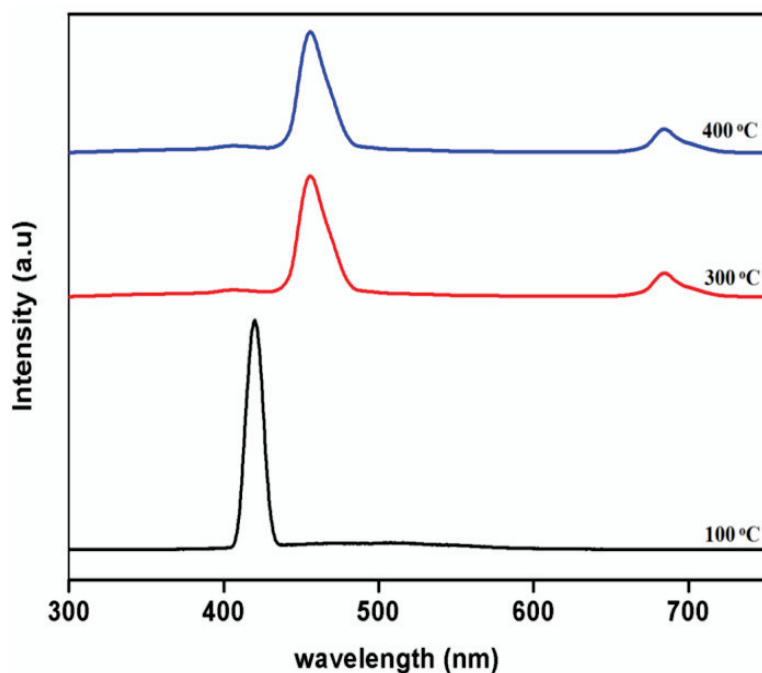


Fig 5.5 Photoluminescence spectrum for the samples annealed at 100°C, 300°C and 400°C.

5.4. Conclusion

The nanoparticles of pure CuI and CuI/CuO mixed phase have been prepared by precipitation method and annealed at four different temperatures. The structural and optical properties of annealed nanoparticles have been examined by various characterization techniques. The XRD patterns show pure γ -CuI phase at 100 and 200 °C while the emergence of additional peaks at higher annealing temperatures represents the formation of monoclinic CuO phase along with γ -CuI. The presence of CuO in the sample annealed at 300 and 400 °C is further confirmed from FT-IR spectral studies and the hydrophobicity of the samples was found to decrease with the formation of mixed phase. CuI and CuI/CuO nanoparticles morphology were studied from SEM micrographs which show the nucleation of CuO nanoparticles with reduction of particle size in the samples annealed at higher temperature. The bandgap value from UV-Vis spectra for the sample annealed at 100 °C matches well with pure CuI and the absorption spectra of CuI nanoparticles were masked by the absorption of the CuO particle formed at higher annealing temperature. The photoluminescence spectra shows optical emission from pure CuI phase at lower annealing temperatures below 200°C whereas additional emission peak due to defect emission was also observed in mixed CuI-CuO phase samples that were annealed at higher annealing temperatures.